RESEARCH PAPER

The Effect of Pre-Reduction of Graphene Oxide on the Electrochemical Performance of rGO-TiO$_2$ Nanocomposite

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Abstract: The graphene oxide -TiO$_2$ (GO-TiO$_2$) and pre-reduced graphene oxide -TiO$_2$ (rGO-TiO$_2$) nanocomposites were fabricated successfully by hydrothermal method. The microstructure of synthesized nanocomposites was investigated using field emission scanning electron microscopy (FESEM) equipped with energy dispersive spectroscopy (EDS) analysis. Moreover, galvanostatic charge/discharge (GCD), cyclic voltammetry (CV), and electrochemical impedance spectroscopy (EIS) methods in three electrode system were applied to evaluate electrochemical properties. The FESEM results clearly showed the formation of TiO$_2$ nanoparticles with average particle size of 25 nm on graphene sheets in both samples. The rGO-TiO$_2$ and GO-TiO$_2$ nanocomposites showed 224 and 32 F/g specific capacitance at 5 mV s^{-1} scan rate in 1 M KOH aqueous electrolyte, respectively. The pre-reduction of graphene oxide is the main reason for the better electrochemical performance of rGO-TiO$_2$ compared to GO-TiO$_2$ nanocomposite samples.

Keywords: Pre-reduction, Nanocomposite, Hydrothermal, Reduced Graphene Oxide, Supercapacitor, TiO$_2$.

1. INTRODUCTION

The ever-expanding need for portable electronic devices such as laptops, mobiles, and electric vehicles, as well as the non-renewability of fossil fuel sources and their environmental pollution, are major reasons that have led to extensive research on renewable energy sources and consequently the development of energy storage devices [1-3]. Electrochemical capacitors (ECs), also known as supercapacitors (SCs), are believed to be one of the most promising and practical energy storage devices, due to their fast charge-discharge rates, high power density (compared with batteries), simple mechanism, and long life cycle [4]. Carbon materials, conducting polymers, and transition metal oxides are three families of the electrode materials for SCs [5, 6]. It is well known that the electrode materials mainly rule the performances of SCs. The energy storage mechanism of transition metal oxides such as MnO$_2$, RuO$_2$, ZnO, and TiO$_2$ are based on the reversible and very fast redox reactions which result in higher charge storage and specific capacity of SCs [5, 7]. Unfortunately, during the redox processes, the structures of these metal oxides can be easily damaged which can result in low electrical conductivity and poor cycling stabilities and restrict the use of these materials as electrodes in the SCs [8, 9].

It seems that the hybridization of transition metal oxides with carbon materials, especially graphene with exceptional thermal, mechanical, and electrical properties, can improve the efficiency of electrode materials by increasing the specific capacity of SCs [10, 11]. Today, the chemical exfoliation method is commonly used to synthesize graphene. In this method, graphite should be converted first to graphene oxide (GO) using strong oxidizing agents, a nonconductive hydrophilic carbon material, afterward, by removing oxygen functional groups, reduced graphene oxide (rGO) can be obtained [11]. Therefore recently graphene based nanocomposites such as rGO-ZnO, rGO-Fe$_2$O$_3$, rGO-RuO$_2$, rGO-MnO$_2$, and rGO-TiO$_2$ have been considered as efficient electrode materials in supercapacitors [10, 12 and 13].

Among these materials, rGO-TiO$_2$ nanocomposite has attracted the attention of many researchers due to low cost, nontoxicity, eco-friendliness, the high specific energy density, and abundant availability of TiO$_2$ [14, 15]. For example, Mishra et al. decorated TiO$_2$ nanoparticles over functionalized graphene by sol-gel method [16]. Xiang et al. used the hydrothermal method to synthesized rGO/TiO$_2$ nanobelts and nanoparticles as SC electrodes. Their results revealed a specific capacitance of 200 and 60Fg^{-1} at a scan rate of 2mVs^{-1}, but only

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56.2 and 18.4F/g at a scan rate of 100mVs^{-1}, for nanobelts and nanoparticles, respectively [17]. Sun et al. used atomic layer deposition to coat graphene sheets with TiO_2 nanoparticles and demonstrated a specific capacitance of 75 Fg^{-1} at a scan rate of 10 mVs^{-1} [18]. Liu et al. have also investigated a GO/TiO_2 nanorod composite to attain a specific capacitance of 100F/g at 5mVs^{-1} scan rate [19].

Reduction of GO make changes in its physical and chemical properties [20], so it appears that the pre-reduction of GO will also affect the electrochemical properties of GO/TiO_2 nanocomposite. The latter point has not been studied by other researchers to the best of authors' knowledge.

In this research, the effect of pre-reduction of graphene oxide before the composite formation process on the quality of the synthesized nanocomposite and its electrochemical performance were examined. For this purpose, GO−TiO_2 and rGO-TiO_2 nanocomposites were synthesized using the hydrothermal process. Reduced graphene oxide, which was reduced for 18 hours, was used as a precursor for rGO-TiO_2 nanocomposite sample synthesis.

2. EXPERIMENTAL PROCEDURE

All chemical reagents with analytical grade used as received in this work (Merck, analytical grade). (For Hummers method: Graphite powder as precursor of GO, potassium permanganate ($KMnO_4$), phosphoric (H_3PO_4), sulfuric acid (H_2SO_4 98%) and hydrogen peroxide as strong oxidizers, ethanol, acetone, deionized (DI) water and hydrochloric acid (HCl) for washing), titanium tetraisopropoxide (TTIP) as precursor of TiO_2, Potassium hydroxide (KOH) as electrolyte and nickel foam.

The graphene oxide (GO) was fabricated using an improved Hummers method [11]. For the synthesis of nanocomposite samples, first 0.4 ml of TTIP was added drop-wise to 10 ml of ethanol under vigorous stirring for 1 hour (IKA Magnetic Stirrers, RCT basic). 8 ml of the resulting solution was added dropwise to the 20 mL of GO suspension (2 mg mL^{-1}) and stirred for 1 hour and was placed in an ultrasonic bath (GRANT_17002) for one hour (volumetric ratio of TTIP: GO suspension was 0.016). Afterwards, it was poured into a 40 ml autoclave and heated at

180 ºC for 12 hours. Finally, the autoclave was allowed to reach room temperature and its contents were washed three times with DI water. It was then dried at 70 ºC in an oven to obtain GO-TiO_2 powder. The same steps were taken to synthesize rGO-TiO_2 nanocomposite samples, except that the reduced graphene oxide suspension, which was reduced for 18 hours via hydrothermal process at 180 °C, was used instead of the graphene oxide suspension. The microstructural study of the GO-TiO_2 and rGO-TiO_2 nanocomposites were conducted using field emission scanning electron microscope (FESEM) equipped with energy dispersive spectroscopy (EDS) analysis (MIRA3 TESCAN-XMU).

Cyclic voltammetry (CV) in a potential window from 0 V to 0.5 V for GO-TiO_2, and 0 V to 0.65 V for rGO-TiO_2 at 5, 10, 30, 50 and 100 mV/s scan rates, galvanostatic charge/discharge (GCD) at current density 1 A g^{-1} in the same potential window, and electrochemical impedance spectroscopy (EIS) at frequency range of 0.1 Hz-100 KHz, with 10 mV AC amplitude, of the GO-TiO_2 and rGO-TiO_2 nanocomposites were all evaluated using Auto-lb (PGSTAT 302N) and three-electrode system including a counter electrode (Pt wire), reference electrode (Ag/AgCl), and a working electrode (Ni foam-coated GO-TiO_2 and rGO-TiO_2 nanocomposites) all three electrodes were immersed in 1 M KOH aqueous solution.

3. RESULTS AND DISCUSSION

The FESEM images of GO-TiO_2 and rGO-TiO_2 nanocomposite samples at different magnifications are shown in Fig. 1. The microstructures clearly reveal the formation of nanoparticles with average particle size of 25 nm on graphene sheets in both samples. However, as can be seen, non-uniformity and concentration of nanoparticles at the surface of the sheets in GO-TiO_2 nanocomposite is higher than those in rGO-TiO_2 nanocomposite. It should be noted that the density of TiO_2 nanoparticles on graphene sheets in GO-TiO_2 nanocomposite is so high that graphene sheets are covered with agglomerated nanoparticles. It seems that the removal of the oxygen functional groups (which act as suitable sites for binding nanoparticles to the graphene sheets) from graphene sheets during the pre-reduction process has led to a decrease in the

concentration of nanoparticles in the rGO-TiO$_2$ nanocomposite.

Elemental map analysis of GO-TiO$_2$ and rGO-TiO$_2$ nanocomposites is shown in Figs. 2a, and Fig. 2b, respectively. These images also reveal the presence of more TiO$_2$ particles on the surface of graphene sheets in nanocomposite GO-TiO$_2$ than in rGO-TiO$_2$ nanocomposite. According to Fig. 2 due to the pre-reduction process in rGO-TiO$_2$ nanocomposite synthesis, the density of oxygen atoms is lower than in GO-TiO$_2$ nanocomposite sample.

Fig. 1. FESEM images of rGO-TiO$_2$ (a, c, and e) and GO-TiO$_2$ (b, d, and f) nanocomposites at different magnifications.

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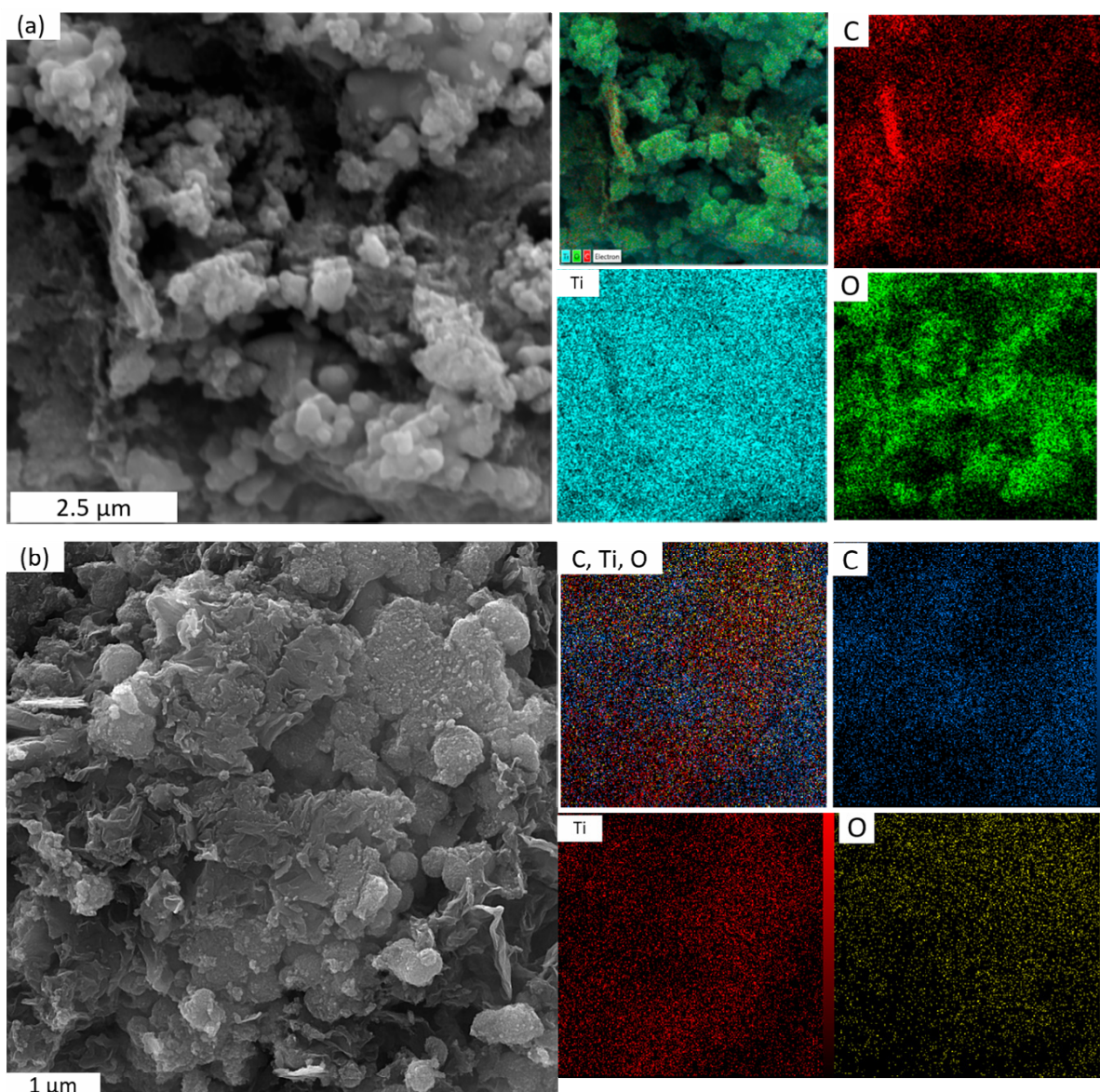*Z. Abasali karaj abad, A. Nemati, A. Malek Khachatourian and M. Golmohammad*

Fig. 2. Elemental map analysis of GO-TiO$_2$ (a) and rGO-TiO$_2$ (b) nanocomposite samples.

In order to evaluate the electrochemical performance of nanocomposites, CV test was performed in a potential window from 0 V to 0.5 V for GO-TiO$_2$, and 0 V to 0.65 V for rGO-TiO$_2$ at different scan rates (Fig. 3a and Fig. 3b respectively). A pair of redox peaks can be observed in the CV curves of rGO-TiO$_2$ nanocomposite (Fig. 3b), which indicates the existence of a pseudocapacitance mechanism in the charge storage [21]. Additionally, the double layer mechanism also might play a minor role in charge storage in this nanocomposite. While in the CV curves of GO-TiO$_2$ nanocomposite (Fig. 3a) no specific peak was observed and the charge storage mechanism was mainly due to the

electrical double layer. For comparison, in Fig. 3c, the CV curves of both nanocomposites are re-drawn at a scan rate of 50 mV s^{-1} in the same plot. The much wider CV curve of the rGO-TiO$_2$ nanocomposite shows the better electrochemical performance of this nanocomposite. The specific capacitance of each electrode was calculated using the following equation (Eq. 1):

$$C = 1/2m\upsilon\Delta V \int i dV \qquad (1)$$

Where, m is the mass (g) of the electrode, υ is the scan rate (V/s), ΔV is the potential window (V), i is the current (A), and $\int i dV$ is the integration area for the CV curves [22]. The specific capacitance values of rGO-TiO$_2$ and

GO-TiO$_2$ nanocomposites were obtained 224 and 32 F/g at 5 mV s^{-1} scan rate, respectively. The results indicated that the pre-reduction of graphene oxide has led to a multiplier increase in the specific capacity.

The GCD curves of rGO-TiO$_2$ and GO-TiO$_2$ nanocomposites at a current density of 1 A g^{-1} is shown in Fig. 4. The shape of the GCD curve of nanocomposites is quite different. The shape of the GCD curve of GO-TiO$_2$ nanocomposite is similar to an isosceles triangle, while the shape of the GCD curve of rGO-TiO$_2$ nanocomposite is out of symmetrical shape and has areas with constant potential which indicates the presence of pseudo-capacitance behavior in this electrode [23]. It is also observed that the charge-discharge time of GO-TiO$_2$ nanocomposite (about 4 s) is much shorter than the charge-discharge time of rGO-TiO$_2$ nanocomposite (about 47 s). Generally, these curves confirm the superiority of rGO-TiO$_2$ nanocomposite electrochemical performance over rGO-TiO$_2$ nanocomposite,

which is in good agreement with CV results.

EIS test was performed (0.1 Hz - 100 KHz, with 10 mV AC amplitude) for rGO-TiO$_2$ and GO-TiO$_2$ nanocomposites, and the results are presented in the form of Nyquist plots in Fig. 5. It can be seen that internal resistance (Rs; determined by the intersection of plot on the left of the real axis) of rGO-TiO$_2$ nanocomposite (1.2Ω) is less than that of the GO-TiO$_2$ nanocomposite (2.5Ω)[24]. Therefore, it can be concluded that the pre-reduction of graphene oxide has led to a decrease in the internal resistance of the rGO-TiO$_2$ nanocomposite. Moreover, the higher slope of the linear part of the rGO-TiO$_2$ nanocomposite in the low frequency region indicates the rapid diffusion of ions and the high rate of double layer formation. According to the microstructure observed in FESEM images, it was predictable that the electrode material of the rGO-TiO$_2$ sample would have more access to electrolyte and faster ion diffusion.
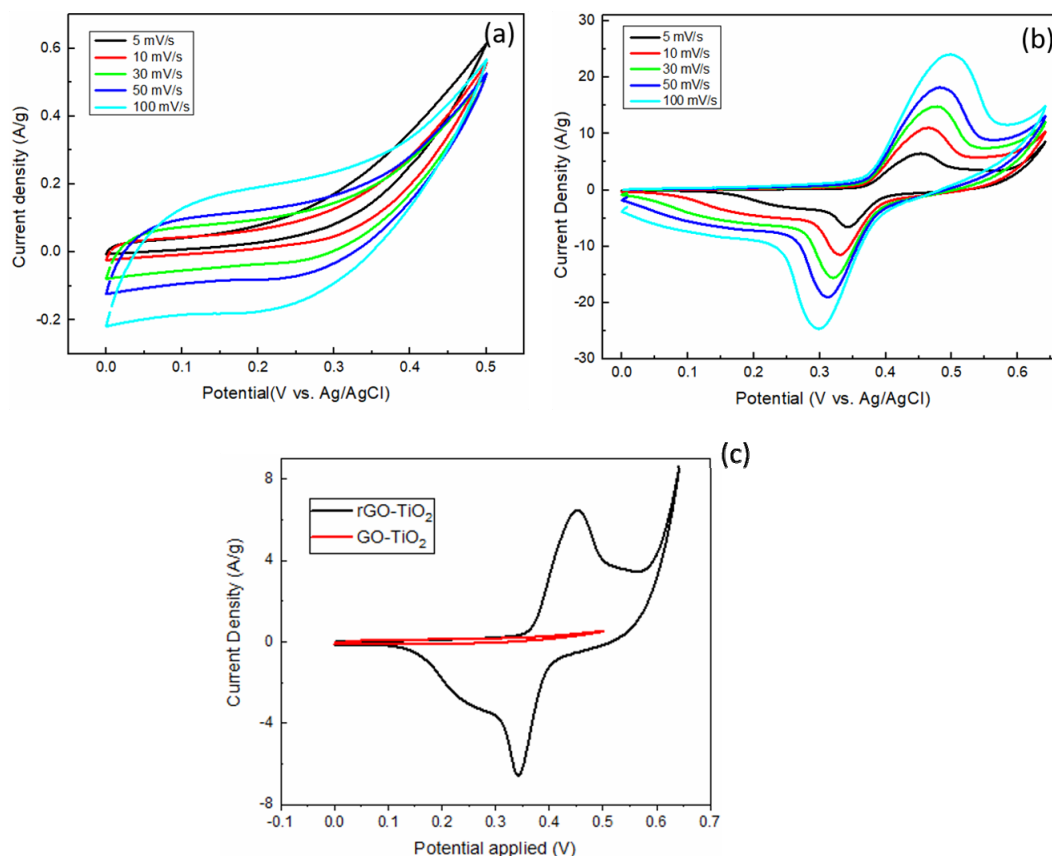

Fig. 3. CV curves of the GO-TiO$_2$ (a) and rGO-TiO$_2$ (b) nanocomposites at 5, 10, 30, 50 and 100 mVs^{-1} scan rate, and GO-TiO$_2$ and rGO-TiO$_2$ nanocomposites at 50 mVs^{-1} scan rate (c).

Fig. 4. GCD curve of rGO-TiO$_2$ and GO-TiO$_2$ nanocomposites at current density 1 A g^{-1}.

Fig. 5. Nyquist plots of EIS of rGO-TiO$_2$ and GO-TiO$_2$ nanocomposite samples

4. CONCLUSION

In this study, the effect of the pre-reduction of graphene oxide on the electrochemical performance of rGO-TiO$_2$ nanocomposite was investigated. FESEM images showed that using the pre-reduction process a more uniform and controlled distribution of nanoparticles on graphene sheets could be achieved. It was demonstrated that the nanocomposite samples made of pre reduced GO has better energy storage performance. The specific capacitance values of rGO-TiO$_2$ and GO-TiO$_2$ nanocomposites were 224 and 32 F/g at 5 mV s^{-1} scan rate from CV curves, respectively. This significant increase in specific capacitance could be due to a decrease in the concentration of TiO$_2$ on graphene sheets, as both electrode materials,

graphene, and TiO$_2$, play an effective role in the energy storage process. Moreover, improvement in the electrochemical performance of rGO-TiO$_2$ nanocomposite electrodes was also observed by a longer charge-discharge process for this sample according to GCD results and lower internal resistance according to EIS results. Based on the results obtained in this study, the pre-reduction of graphene oxide can be used as an effective solution to increase the electrochemical efficiency of GO-TiO$_2$ nanocomposite samples.

5. REFERENCES

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